

Ground-state tuning of metal-insulator transition by compositional variations in $\text{BaIr}_{1-x}\text{Ru}_x\text{O}_3$ ($0 \leq x \leq 1$)

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Hexagonal BaIrO_3 is a magnetic insulator driven by the spin-orbit interaction (SOI), whereas BaRuO_3 is an enhanced paramagnetic metal. Our investigation of structural, magnetic, transport, and thermal properties reveals that substitution of Ru^{4+} ($4d^4$) ions for Ir^{4+} ($5d^5$) ions in BaIrO_3 reduces the magnitudes of the SOI and a monoclinic structural distortion and rebalances the competition between the SOI and the lattice degrees of freedom to render an evolution from a magnetic insulating state to a robust metallic state. The central findings of this paper are as follows: (1) light Ru doping ($0 < x \leq 0.15$) prompts simultaneous, precipitous drops in both the magnetic ordering temperature T_N and the electrical resistivity, and (2) heavier Ru doping ($0.41 \leq x \leq 0.9$) induces a robust metallic state without any long-range magnetic order. All results suggest a critical role of the lattice degrees of freedom in determining the ground state in the heavy transition-metal oxides.

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I. INTRODUCTION

A unique feature of the $5d$ iridates is that a strong spin-orbit interaction (SOI) competes vigorously with Coulomb interactions, noncubic crystalline electric fields, and Hund's rule coupling [1–5]. The relative strengths of these interactions stabilize new exotic ground states that provide a fertile ground for studying new physics. In particular, it is now recognized that strong SOI can drive novel narrow-gap Mott insulating states in iridates. The SOI is a relativistic effect that is proportional to Z^2 (Z is the atomic number), is approximately 0.4 eV in the iridates (compared to ~ 20 meV in $3d$ materials), and splits the t_{2g} bands into states with $J_{\text{eff}} = 1/2$ and $J_{\text{eff}} = 3/2$, the latter having lower energy. Since the Ir^{4+} ($5d^5$) ions provide five $5d$ valence electrons, four of them fill the lower $J_{\text{eff}} = 3/2$ bands, and one electron partially occupies the $J_{\text{eff}} = 1/2$ band in which the Fermi level E_F resides. The $J_{\text{eff}} = 1/2$ band is so narrow that even a reduced U (~ 0.50 eV due to the extended nature of $5d$ -electron orbitals) is sufficient to open a gap (≤ 0.62 eV) that induces a novel insulating state, which is contrary to expectations based upon the relatively large unsplit $5d$ bandwidth [1–3,6].

Adopting a distorted hexagonal structure with both face-sharing and corner-sharing IrO_6 octahedra, BaIrO_3 is particularly unique in that it exhibits a simultaneous onset of weak ferromagnetic transition due to a canted antiferromagnetic structure and charge-density wave (CDW) orders with Néel temperature $T_N = 183$ K, comparable to that of other iridates, such as 240 K for Sr_2IrO_4 [7] and 285 K for $\text{Sr}_3\text{Ir}_2\text{O}_7$ [8], and a temperature-driven transition from a bad metal to an insulating ground state [9–11]. The ground state of BaIrO_3 is extremely sensitive to lattice contractions that can be tuned by light doping or the application of hydrostatic pressures [4,12,13]. The extraordinary delicacy of the ground state in BaIrO_3 implies a critical balance among orbital, electronic, and lattice

degrees of freedom [4,14]. The hexagonal structure of BaIrO_3 is similar to that of nine-layered rhombohedral BaRuO_3 , which exhibits a crossover from metallic to insulating behavior and enhanced paramagnetism with decreasing temperature [15,16]. However, a monoclinic distortion extant in BaIrO_3 at room temperature and 90 K generates twisting and buckling of the cluster trimers (see Fig. 1) that give rise to two one-dimensional (1D) zigzag chains along the c axis and a two-dimensional layer of corner-sharing IrO_6 octahedra on the ab plane [9,12,17–19].

Although BaIrO_3 and BaRuO_3 have similar structures, they exhibit sharply contrasting physical properties, which underscore the critical role SOI (~ 0.4 eV for iridates and ~ 0.15 eV for ruthenates) [3], and the lattice degrees of freedom can play in determining the ground state in iridates. In this paper, substituting Ru^{4+} ($4d^4$) for Ir^{4+} ($5d^5$) in single-crystal $\text{BaIr}_{1-x}\text{Ru}_x\text{O}_3$ ($0 \leq x \leq 1$) reduces the magnitude of the SOI, the structural distortions, and adds holes to the t_{2g} bands. The overall effect of Ru doping is to lower E_F and move the system away from the Mott instability toward a more robust metallic state. The emerging metallic state with delocalized electrons also accompanies a decrease in T_N .

II. EXPERIMENT

The single crystals of $\text{BaIr}_{1-x}\text{Ru}_x\text{O}_3$ were grown by conventional flux methods similar to earlier reports [9,15] using BaCl_2 as a self-flux. Crystals were grown in platinum crucibles using IrO_2 (99.98%, Alfa Aesar), RuO_2 (99.98%, Alfa Aesar), BaCO_3 (99.99%, Alfa Aesar), and anhydrous BaCl_2 (99.5%, Alfa Aesar). Starting powders were placed in a Pt crucible with a Pt lid, and this assembly was then put in an alumina crucible with a cover. The mixtures were heated up to 1480 °C and then cooled to 1350 °C at a rate of 5 °C per hour before cooling down to room temperature. The ratio of the sample to flux remains at 1:8 throughout the entire series of $\text{BaIr}_{1-x}\text{Ru}_x\text{O}_3$. The crystals have a hexagonal surface and a visible layered texture along the c axis as

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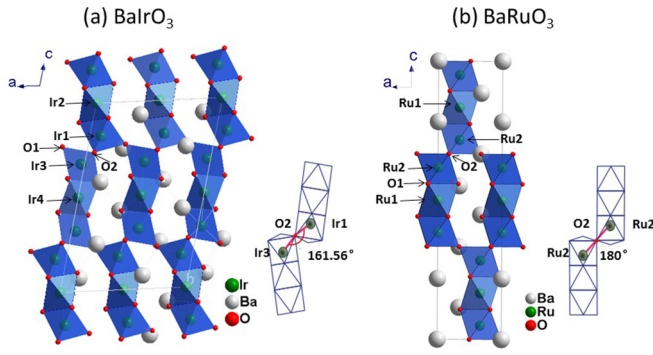


FIG. 1. Comparison of the nine-layer crystallographic form (a) BaIrO_3 and (b) BaRuO_3 crystal structure. Note the corner-sharing Ir_3O_{12} and Ru_3O_{12} trimers that are connected through the vertices of the top and bottom octahedra of the trimers and the schematic of the $M\text{-O}_2\text{-M}$ bond angle θ ($M = \text{Ir}$ or Ru).

shown in the inset of Fig. 2(a). The crystal structures were determined using a Nonius Kappa CCD x-ray diffractometer or a Rigaku x-ray diffractometer XtaLAB PRO equipped with a PILATUS 200-K hybrid pixel array detector at 90 or 240 K, and they were refined by full matrix least squares using the SHELX-97 programs [20]. The standard deviations of all lattice parameters and interatomic distances are smaller than 0.1%. The atomic parameters for $\text{BaIr}_{1-x}\text{Ru}_x\text{O}_3$ are available in the Supplemental Material [21]. Chemical compositions of the single crystals were estimated using a combined unit of

Hitachi/Oxford SwiftED 3000 for energy dispersive x-ray spectroscopy. The magnetization $M(T)$, electrical resistivity $\rho(T)$, and specific heat $C(T)$ were measured between 1.7 and 400 K using a Quantum Design 7T superconducting quantum interference device (SQUID) magnetometer and a Quantum Design 9T physical property measurement system, respectively.

III. RESULTS AND DISCUSSION

The two end members BaIrO_3 and BaRuO_3 both have nine-layer rhombohedral phases with different space groups as shown in Figs. 1(a) and 1(b). The $C2/m(12)$ space group of BaIrO_3 features three face-sharing IrO_6 octahedra forming Ir_3O_{12} trimers that are corner and face shared via IrO_6 octahedra (containing Ir_1 and Ir_3 sites) to form 1D chains along the c axis [12,16–19] [see Fig. 1(a)]. A monoclinic distortion generates twisting and buckling of the trimers (tilted $\sim 12^\circ$ relative to each other), which gives rise to two 1D zigzag chains along the c axis and a two-dimensional layer of corner-sharing IrO_6 octahedra on the ab plane. Substituting Ru^{4+} for Ir^{4+} preserves the monoclinic structure in the entire doping range ($x \leq 0.90$) except for $x = 1$ as shown in Table I. It results in a nearly uniform reduction in lattice parameters a - c axes and the unit-cell volume V . This behavior is expected because the ionic radius of Ru^{4+} (0.620 Å) is slightly smaller than that of Ir^{4+} (0.625 Å). In addition, the Ir/Ru-O-Ir/Ru bond angle θ increases linearly with increasing Ru concentration x and eventually reaches 180° for $x = 1$ (i.e., BaRuO_3), indicating a significantly less distorted lattice. BaRuO_3 or $x = 1$ exhibits a similar crystal structure with the $R\bar{3}m(166)$ space group as shown in Fig. 1(b). Three RuO_6 octahedra share faces in a partial chain, facilitating direct Ru-Ru d -orbital interactions between the octahedra. Each of these triple units or trimers of the octahedra shares corners with its neighbors along the hexagonal axis via nearly 180° bond angles that favor superexchange coupling [Fig. 1(b)].

Ru doping induces pronounced changes in a wide range of physical properties of single-crystal $\text{BaIr}_{1-x}\text{Ru}_x\text{O}_3$. Representative data for the c -axis magnetic susceptibility $\chi_c(T)$ that shows the weak magnetic transition at T_N is depressed from 183 K for $x = 0$ to 145 K for $x = 0.04$ and vanishes for $x \geq 0.41$ is presented in Fig. 2.

The magnetic anisotropy also decreases with Ru additions as shown in Fig. 3. Magnetic anisotropy is in general a result of SOI; Ru doping weakens the SOI, therefore, leading to a smaller magnetic anisotropy. Furthermore, Hund's rule coupling competes with the SOI and thus weakens the relative strength of the SOI. With increasing x , the c -axis susceptibility $\chi_c(T)$ becomes relatively stronger and larger than the basal-plane susceptibility $\chi_{ab}(T)$ [see Figs. 3(b) and 3(c)]. This change suggests a spin flop from the basal plane to the c axis due to Ru doping. For $x = 1$, the basal-plane $\chi_{ab}(T)$ is larger than $\chi_c(T)$ again [see Fig. 3(d)]. Similar phenomena were also observed in $\text{Ca}_2\text{Ru}_{1-x}\text{Ir}_x\text{O}_4$ [22] and $\text{Sr}_2\text{Ir}_{1-x}\text{Ru}_x\text{O}_4$ [23]. This behavior could be due to the strong interaction between Ru $4d$ and Ir $5d$ electrons.

It is already established that the bond angle θ is critical to the electronic and magnetic structures of iridates [4]. As shown in Fig. 4(a), θ increases linearly with increasing x and

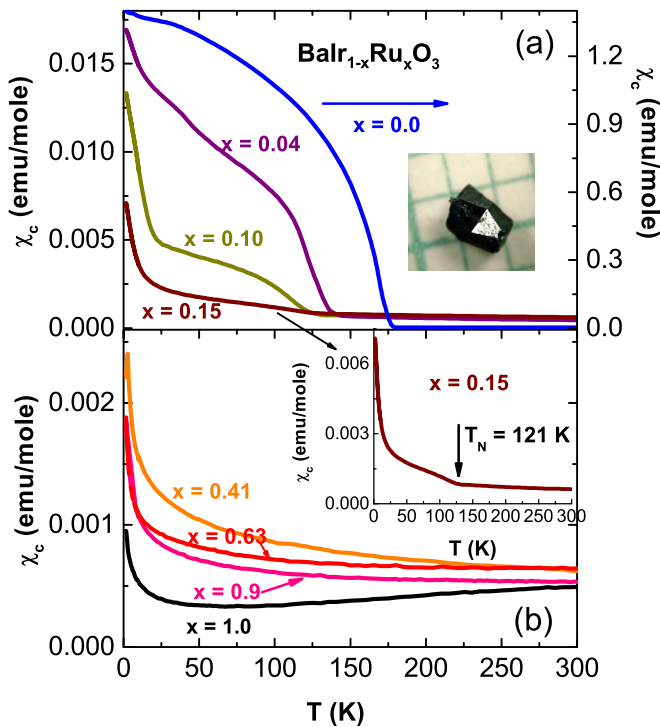


FIG. 2. The magnetic susceptibilities $\chi(T)$ along the c axis for $\text{BaIr}_{1-x}\text{Ru}_x\text{O}_3$ where (a) $0 \leq x \leq 0.15$ and (b) $0.42 \leq x \leq 1$. The data were collected after a field cooling procedure at $\mu_0 H = 0.1$ T. The inset in (a) shows a representative single crystal of $\text{BaIr}_{1-x}\text{Ru}_x\text{O}_3$ with $x = 0$. The inset in (b) shows an enlarged $\chi_c(T)$ for $x = 0.15$.

TABLE I. The crystal structure and refinement details of $\text{BaIr}_{1-x}\text{Ru}_x\text{O}_3$ at 90 K for $x = 0, 0.10, 0.63$, and 1 and at 240 K for $x = 0.82$ and 0.90. The diffractometer is a Nonius Kappa CCD, and the absorption correction is a multiscan SADABS. The Ir/Ru-O₂-Ir/Ru bond angle is defined in Fig. 1.

	$x = 0$ (90 K)	$x = 0.10$ (90 K)	$x = 0.63$ (90 K)	$x = 0.82$ (240 K)	$x = 0.90$ (240 K)	$x = 1$ (90 K)
Crystal system, space group	Monoclinic, $C12/m1(12)$	Monoclinic, $C12/m1(12)$	Monoclinic, $C12/m1(12)$	Monoclinic, $C12/m1(12)$	Monoclinic, $C12/m1(12)$	Trigonal, $R\bar{3}m(166)$
a - c (Å)	$a = 9.9935(2),$ $b = 5.7352(1),$ $c = 15.2376(3)$	$a = 9.9839(2),$ $b = 5.7377(1),$ $c = 15.1107(4)$	$a = 9.9440(2),$ $b = 5.7429(1),$ $c = 14.8102(4)$	$a = 9.9999(5),$ $b = 5.7759(4),$ $c = 14.8916(4)$	$a = 9.9923(4),$ $b = 5.7733(3),$ $c = 14.8882(8)$	$a = 5.7366(1),$ $c = 21.5933(6)$
β (deg)	103.411(1)	103.3402(9)	102.8574(9)	102.939(5)	102.882(4)	NA
V (Å ³)	849.10(6)	842.25(3)	824.57(3)	838.28(8)	837.26(7)	615.40(3)
Z	12	12	12	12	12	9
Bond angle (deg)	161.671(1)	163.678(0)	174.296(1)	175.1(3)	176.1(1)	180.0
	Data collection					
Number of measured, independent, and observed [$I > 4\sigma(I)$] reflections	6066,398,350	7075,396,369	7210,398,353	14459,1643,1525	14071,1769,1633	7256,401,398
R_{int}	0.021	0.031	0.035	0.027	0.038	0.025
	Refinement					
$R[F^2 > 4\sigma(F^2)],$ $wR(F^2), S$	0.016,0.035,1.05	0.02,0.049,1.15	0.025,0.069,1.17	0.067,0.1847,1.085	0.0720,205,1.024	0.02,0.035,1.09

eventually reaches an ideal 180° for $x = 1$. The increase in θ directly enhances the electron hopping and favors a more metallic state with a concurrent decrease in T_N [see Fig. 4(b)].

Indeed, the evolution from the insulating to the itinerant state upon Ru doping is clearly illustrated in the electrical resistivity $\rho(T)$. For $x = 0$, both the ab plane and the c axis $\rho_{ab}(T)$ and $\rho_c(T)$ exhibit a sharp kink at $T_N = 183$ K, consistent with previous results in which the energy gap is estimated to be 0.1 eV [9,17]. With Ru doping, both $\rho_{ab}(T)$ and $\rho_c(T)$ decrease rapidly (see Fig. 5). It is noted that the metallic

behavior at higher temperatures for $x = 0.04$ [see Fig. 5(b)] does not seem to follow the general trend displayed by other compositions although the behavior is highly reproducible.

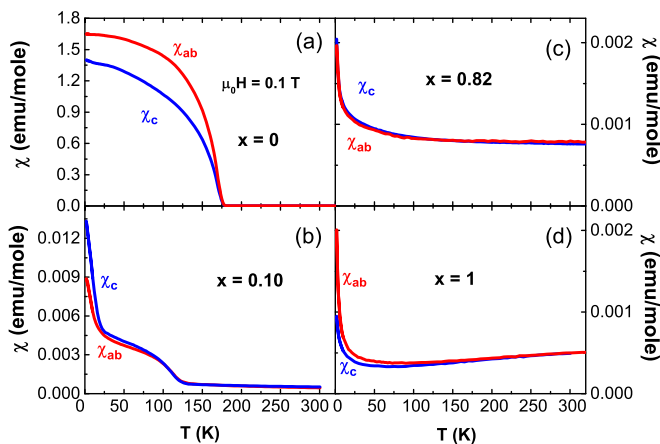


FIG. 3. The magnetic susceptibilities $\chi(T)$ on the ab plane and along the c axis for representative compositions (a) $x = 0$, (b) $x = 0.10$, (c) $x = 0.82$, and (d) $x = 1$, respectively. The magnetization was measured after field cooling at $\mu_0 H = 0.1$ T.

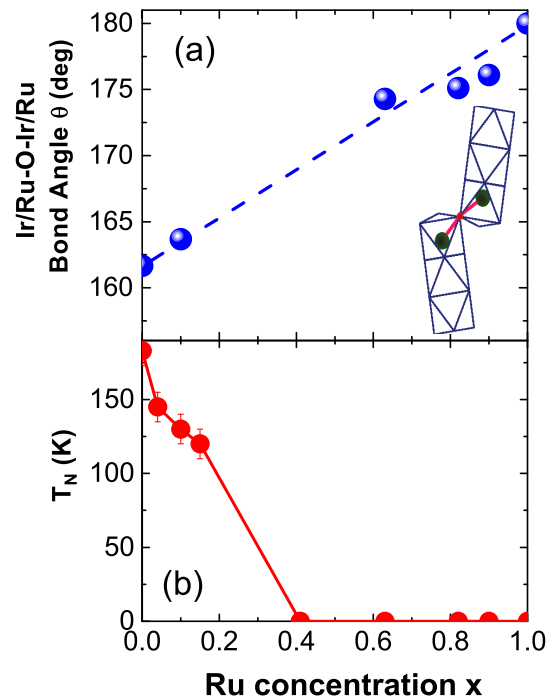


FIG. 4. The Ru concentration x dependence of (a) the Ir/Ru-O₂-Ir/Ru bond angle θ and (b) T_N . The inset: schematic of the Ir/Ru-O₂-Ir/Ru bond angle θ . Note that θ increases linearly with increasing x .

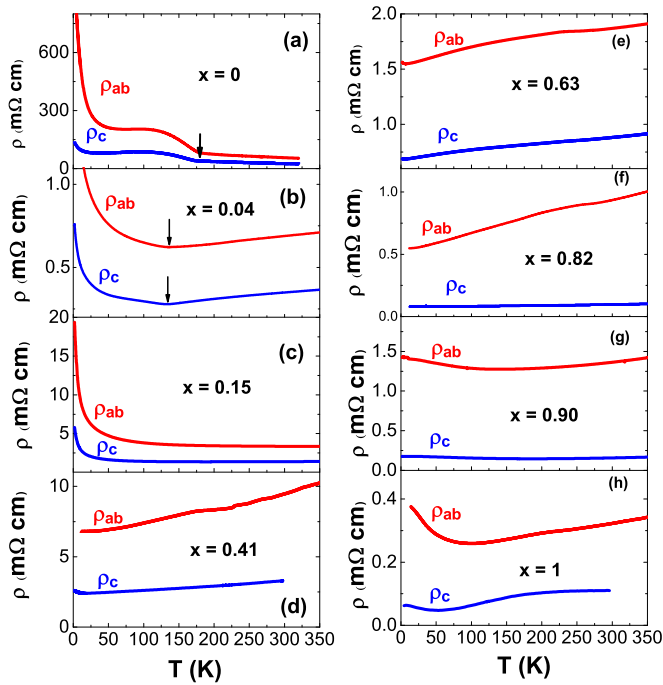


FIG. 5. The temperature dependence of the resistivity $\rho(T)$ for representative compositions (a) $x = 0$, (b) $x = 0.04$, (c) $x = 0.15$, (d) $x = 0.41$, (e) $x = 0.63$, (f) $x = 0.82$, (g) $x = 0.90$, and (h) $x = 1$. The vertical arrows indicate the kink that corresponds to the weak magnetic transition at $T = T_N$.

The origin of this brief occurrence of the metallic state is yet to be understood. Nevertheless, dilute Ru substitutions for Ir result in a reduced $\rho(T)$ and an emerging metallic state for $x > 0.15$. For $x = 1$ or BaRuO_3 , a broad upturn in $\rho_{ab}(T)$ at low temperatures might be a result of a pseudogap formation and 1D-CDW fluctuations according to Ref. [16].

The temperature dependence of the specific heat $C(T)$ for various x 's is given in Fig. 6(a). Fitting the data to $C(T) = \gamma T + \beta T^3$ for $7 < T < 17$ K yields the Sommerfeld coefficient γ for the electronic contribution to $C(T)$ [see Fig. 6(b)], which serves as a measure of the electronic density of states at the Fermi level $N(E_F)$ and the effective mass of the carriers. There is a substantial increase in γ with dilute Ru concentration; in particular, γ reaches $11.75 \text{ mJ mol}^{-1} \text{ K}^{-2}$ for $x = 0.04$ and $15.09 \text{ mJ mol}^{-1} \text{ K}^{-2}$ for $x = 0.15$, compared to $\gamma = 2.34 \text{ mJ mol}^{-1} \text{ K}^{-2}$ for the parent compound ($x = 0.0$). The γ for $0.04 \leq x \leq 0.15$ in which the metallic state is not fully developed is unexpectedly high, and this is likely due to spin fluctuations existent in the system. Nevertheless, $N(E_F)$ and γ eventually decrease with x as shown in Fig. 6(b). In the case of BaRuO_3 , the smaller values reflect pseudogap formation due to the CDW instability [16].

IV. CONCLUSIONS

We have investigated the structural, magnetic, transport, and thermal properties of $\text{BaIr}_{1-x}\text{Ru}_x\text{O}_3$. Ru doping rebalances

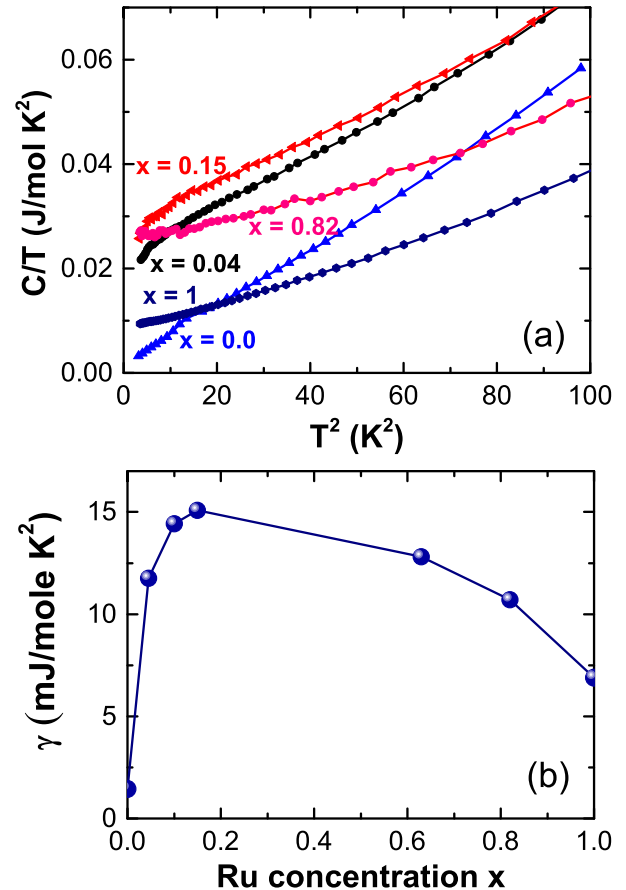


FIG. 6. (a) The specific heat $C(T)/T$ vs T^2 and (b) the Sommerfeld coefficient γ vs x for $\text{BaIr}_{1-x}\text{Ru}_x\text{O}_3$.

the competition among the SOI, electron correlations, and the lattice degrees of freedom to generate a metallic state for $x > 0.15$. The Ru doping alters the relative strength of the SOI that dictates the ground state, which, in turn, affects the band gap near E_F . Unlike the situation in Sr_2IrO_4 that features an unconventional correlation between the magnetic transition and the charge gap, the evolution of the ground state in $\text{BaIr}_{1-x}\text{Ru}_x\text{O}_3$ appears to indicate a strong coupling between the magnetic order and the metal-insulator transition. All results suggest the critical role of lattice degrees of freedom that, along with the SOI, dictate the ground state of the heavy transition-metal oxides.

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